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### ARKIV FÖR KEMI Band 5 nr 4

Communicated 14 May 1952 by ARNE WESTGREN

## The structure of Bi<sub>2</sub>NbO<sub>5</sub>F and isomorphous compounds

By BENGT AURIVILLIUS

With 1 figure in the text

X-ray studies made previously on the compounds CaBi<sub>2</sub>Nb<sub>2</sub>O<sub>2</sub>, Bi<sub>4</sub>Ti<sub>3</sub>O<sub>12</sub> and BaBiaTiaO15 (1) showed that they have very similar structures. The symmetry is tetragonal or pseudotetragonal, and the structures are each built up of quadratic BiaO, layers alternating with perovskite layers, the latter having different heights in the three different cases. The generalized formula for the compounds might be written  $Me_2O_2$  ( $Me_{m-1}R_mO_{8m+1}$ ) where Me' is the 12 coordinated metal atom in the perovskite layers and R the 6 coordinated atom. The formulae for the above compounds,  $Bi_2O_2(CaNb_2O_7)$ ,  $Bi_2O_2(Bi_2Ti_3O_{10})$  and  $Bi_2O_2[(BaBi_2) Ti_4O_{13}]$  thus have mvalues of 2, 3 and 4. The simplest case, m=1, would correspond to the formula  $Me_3O_3(RO_4)$  and to a structure built up of  $Me_2O_2$  layers and layers of  $RO_6$  octahedra each octahedron sharing four corners. Compounds of this type have, however, not been successfully synthesized as yet.

The present paper deals with the compounds Bi<sub>2</sub>NbO<sub>5</sub>F, Bi<sub>2</sub>TaO<sub>5</sub>F and Bi<sub>2</sub>TiO<sub>4</sub>F<sub>2</sub>, which correspond to the simplest case, m=1 above, except that some of the O atoms are replaced by F atoms. The formulae of the compounds might thus be written:

 $Bi_2(O, F)_2Nb(O, F)_4$  etc.

### Preparation, powder photographs and analyses

Bi<sub>2</sub>NbO<sub>5</sub>F: When a mixture of BiF<sub>3</sub> and Nb<sub>2</sub>O<sub>5</sub> in the mol ratio 4:1 was heated in air at 800° C for a short time, a few single crystals (very thin plates) were obtained. Powder photographs of this sample indicated a tetragonal unit cell with the same cell dimensions as would be expected for the above general type when m=1. The best conditions for the formation of this phase were then studied by heating 2.5 g mixtures (2BiF<sub>8</sub> + 1/2 Nb<sub>2</sub>O<sub>5</sub>) in air at 640° C, this low temperature being chosen to reduce the volatility of the BiF<sub>3</sub>. The reaction times were varied from 5-40 hours, and powder photographs were taken of each product. For reaction times of 7-15 hours the lines of the above tetragonal phase predominated in the powder photographs, the few extra lines were very weak (see Table 1).

The fluorine content was found to vary from 4.0 % (7 hours) to 2.4 % (15 hours), whereas the calculated value for Bi<sub>2</sub>NbO<sub>5</sub>F is 3.2 %. No variation in the size of the cell with the fluorine content was found, and it therefore seems probable that the composition of the phase is constant and that the observed variation in the F content is due to the presence of small impurities which are not visible in the powder photo-

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Table I

Powder photographs of  $Bi_8NbO_6F$  (sample with 2.8 % F)

Or K radiation  $\lambda_{CrK_a} = 2.2909$  Å

1319 1708 1784 1974 2078 2543 3036 3492 3568 3758 4734 4620	104 · sin <sup>3</sup> θ <sub>obs</sub> 1332 1711 1796 1969 2083  2547 2785 3033 3093 3490 3567 3764 3808 4737	st m m vvw vw st st vw vw st
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3492 3568 3758 	3093 3490 3567 3754 3808	w vw st st vw vw
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3568 3758 	3567 3754 3808	st st vw vw
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4734 4743	3808	vw vw
4743 }		vw
4743 }	4737	1
	4737	st
1000 1		
	4827	w
1888	4891	st
5276	5292	m
5647	5658	W
5527	6529	VW
631	6630	1
830	6841	W
137	7139	w
327	7330	w
076	8080	VW
303 )		VVW
311 }	8304	at
456	8456	m
RIA !	8611	m
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ı	614 845	845 8844 908 \

graphs. To check that the Bi/Nb ratio had not changed essentially during the heating, a complete analysis was made on one sample (9 hours heating). The following values were obtained as the mean of three analyses: Bi:  $68.0 \pm 0.6\%$ , Nb:  $14.2 \pm 0.6\%$ , F: 3.7 + 0.2%.

The values calculated for Bi<sub>2</sub>NbO<sub>5</sub>F are:

Bi: 68.5, Nb: 15.2%, F: 3.2%.

From these figures it seems probable that the formula of the compound is Bi<sub>2</sub>NbO<sub>5</sub>F.

Bi<sub>2</sub>TaO<sub>5</sub>F was prepared in exactly the same way as Bi<sub>2</sub>NbO<sub>5</sub>F. No analysis was made since the powder photographs were similar to those of Bi<sub>2</sub>NbO<sub>5</sub>F and niobium and tantalum compounds are usually isomorphous.

Fig. 1. C

Bi<sub>2</sub>Ti(
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Bi<sub>2</sub>TiO<sub>4</sub>F<sub>2</sub>

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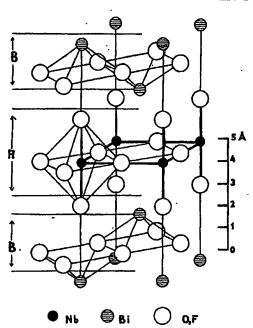


Fig. 1. One half of the unit cell of Bi\_NbO\_F. A denotes the region of Nb (O, F)4 octahedra and B the Bi<sub>2</sub> (O, F)<sub>2</sub> layers.

Bi<sub>2</sub>TiO<sub>4</sub>F<sub>2</sub>: Mixtures of bismuth fluoride and titanium oxide in the mol ratio 2:1 (total 2.3 g) were heated in air at 640° for various lengths of thime. The powder photographs obtained from samples heated for 3 or 5 hours could be interpreted by assuming a mixture of BiOF (2) and a phase whose cell dimensions were nearly the same as those of Bi<sub>2</sub>NbO<sub>5</sub>F. (See Table 2, where the reflexions from the BiOF phase are designated by b and those from the other phase by a). The fluorine contents of the mixed samples were found to be 6.4 % (5 hours) and 8.4 % (3 hours) but no variation of the cell size with the fluorine content was found. The calculated values are 7.8 % for BiOF and 6.7 % for Bi<sub>2</sub>TiO<sub>4</sub>F<sub>2</sub>. From the original Bi/Ti ratio, from the fluorine analysis and from the similarity of the powder photographs with those of  $Bi_2NbO_5F$  (Tables 1 and 2), it was concluded that the formula of the phase is Bi<sub>2</sub>TiO<sub>4</sub>F<sub>2</sub>.

Methods of analysis: Fluorine. The samples were first decomposed by fusing with NaOH, and were then distilled with  $\mathrm{HClO_4}$  as described by Willard and Winter (3). The distillate was titrated with  $Th(NO_8)_4$  using Na-alizarinsulphonate as indicator.

Bismuth and Niobium. The samples were brought into solution, and niobium was determined as described in Scott's "Standard Methods" (4). Bismuth was first precipitated as Bi<sub>2</sub>S<sub>3</sub>, which was then redissolved, converted to Bi<sub>2</sub>O<sub>3</sub> and weighed as such.

### Unit cells and space group

The dimensions of the unit cells were determined from powder photographs taken with focusing cameras of the Phragmén type (Tables 1 and 2). The radiation used

the heating, owing values 2±0.6%, F:

F)

s Bi<sub>2</sub>NbO<sub>6</sub>F.

No analysis NbO<sub>5</sub>F and

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Table 2
Powder photographs of  $Bi_3TiO_4F_2$  (mixed with BiOF) CrK radiation ( $\lambda_{CrK_a}=2.2909$  Å). a denotes the  $Bi_2TiO_4F_2$  phase and b the BiOF phase

			ores rue mail	Pars busse su	d a rue BIO
	h k l	phase	104 sin² θ <sub>oalo</sub>	104 sin² θobs	Iobs
	101	a	957	941	
	101	ь	1270	1265	vvw
	103	a	1351 )	1	m
	1002	ь	1352	1354	st
	006	а	1771	1767	m
	110	, a	1816	1811	m
	110	b	1864	1858	VVW
	112	} a	2013	1999	W
	105	1 0	2138	2132	w+
	102	ь	2284	2288	VVW
	114	a	2603	2601	m
	1.0	-		3074	VW
	112	b	3216	3231	w
	107 116	a	3319	3320	vw
-	200	a	3587	3600	st
ı	200	<i>a</i>	3631	3624	at
1	202	b	3728	3744	∫ w
ĺ	103	a i b	3828	3822	w
i		0	3974	3982	w
	204	a	4470	4352	VVW
Ì	211	a	4418	4426	vvw
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!_	202	6	5080	5084	st w
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	10 11	a	6779 }	i	m
	217	a	6860	6869	m
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	20 10	a	8550 }	8524	st
	303	a	8613	8603	m
	222	b	8736	8725	w
	1 1 12	<i>b</i>	8814	8807	w
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Bi<sub>2</sub>N unit ce the ma: for l od the Pat sity val here) sh of other position or 000. gave the No de For B meter. A occupy t was obta

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Table 2 (continued)

hkl	phase	$10^4 \sin^2  heta_{ m calc}$	104 sin <sup>2</sup> θ <sub>obs</sub>	$I_{ m Obs}$
226	a	9033	9027	m
310	a	9078	9072	w
204	ь	9136	9136	vw
312	a	9275	9272	m
310	ь	9319	9319	W
{ 105 305	b	9382 \ 9400 }	9384	220

 $_{\text{W8S}}$  Cr K ( $\lambda_{\text{CrK}_a} = 2.2909$  Å). As mentioned above the powder photographs could he interpreted by assuming tetragonal unit cells; the cell dimensions are given below.

•	a (Å)	c (Å)
Bi <sub>2</sub> NbO <sub>5</sub> F	. 3.835	16.63
Bi <sub>2</sub> TaO <sub>5</sub> F		16.64
Bi.TiO.F.		16.33

The errors in these figures are estimated to be  $\pm 0.1$  %.

The observed density was 8.0 for Bi<sub>2</sub>NbO<sub>5</sub>F (preparation with 2.8% F), which agrees fairly well with the assumption of 2 formula units per unit cell, giving a calculated density of 8.26.

Zero and first order Weissenberg photographs around the a axis were taken. There was nothing in the Weissenberg photographs to indicate a Laue symmetry lower than  $D_{4h}$ —4/mmm. The only extinctions found were that h, k, l were absent for h+k+l odd, which is characteristic of the space groups  $C_{4\nu}^{0}$ ,  $D_{2d}^{0}$   $D_{2d}^{11}$  and  $D_{4h}^{17}$ .

Powder photographs only were taken of the compounds Bi<sub>2</sub>TaO<sub>5</sub>F and Bi<sub>2</sub>TiO<sub>4</sub>F<sub>2</sub>, and from these it was concluded that these substances are isomorphous with  $\mathrm{Bi_2NbO_6F}$ .

### Positions of the metal atoms

BlaNbOsF: With 2 formula units per unit cell there are 4 Bi and 2 Nb atoms per unit cell. The intensities of the spots in the Weissenberg photographs seemed in the main to depend only on the l values. Thus for l even  $I_{00l} \approx I_{11l} \approx I_{20l}$  etc. and for l odd:  $I_{10l} \approx I_{21l} \approx I_{80l}$  etc. With these intensity values a good approximation of the Patterson-Harker function along 00z could be obtained by using only the intensity values of h0l and h1l. The Patterson-Harker function thus calculated (not given here) showed only one, big, maximum, at z=0.34. This maximum, and the absence of others, could be explained only by assuming that 4 Bi atoms are situated at the positions  $\pm 0.0z$  with z = 0.17 or z = 0.33, and the Nb atoms at the positions  $0.0\frac{1}{2}$ or 000. Arbitrarily choosing 000 as the position for Nb, trial and error calculations gave the value 0.325 for the Bi parameter.

No determination of  $z_{Bi}$  was made for  $Bi_2TaO_5F$ .

For Bi<sub>2</sub>TiO<sub>4</sub>F<sub>2</sub> the powder photograph data were used to determine the Bi parameter. Assuming the Ti atoms to be situated at 000 and the O and the F atoms to occupy the same positions as given below for Bi<sub>2</sub>NbO<sub>5</sub>F, the value  $z_{Bi} = 0.327 \pm 0.006$ was obtained from trial and error calculations.

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Weissenberg photographs of

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Zero layer

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	h k	- <del> </del>	0 0	-	2 0	4	0
1 0 2 4 6 8		I <sub>calc</sub> — 0.3 220 12	_	I <sub>calc</sub> 170 10 0.04 67	Iobs vst w 4 — w vw	I <sub>calc</sub> 37 3 0.05 34 6	Iobs  vw  vw  vw
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1 2 1 4 1 6 1 8 2 0		15 10 16 22	st m w m	26 11 10 28	W W TO	86	st
	h k	1	0	3	0	***************************************	
1 3 5 7 9		I <sub>calo</sub> 7 270 40 1 75	Iobs w vst w m	I <sub>calo</sub> 1 42 8 0.2 30	Iobs vw w vw 		
11		18	w	10	vw		
13 15 17 19	•	3 20 12 14	vw m w	3 19	vw m		

# Positions of the O and the F atoms

The positions of the O and the F atoms could not be distinguished either from the diffraction data, or from space considerations because of the similarity in the reflecting power and ionic radii of O<sup>2-</sup> and F<sup>-</sup>. The problem is therefore treated as though O and F were the same atomic species. In the following, the O and the F atoms are denoted by (O, F) and the discussion relates to Bi<sub>2</sub>NbO<sub>5</sub>F for which z<sub>Bi</sub> could be accurately determined from the Weissenberg photographs.

It seemed reasonable to assume that the Nb atoms are surrounded by a regular or nearly regular octahedron of (O, F) atoms with distances Nb- $(O, F) \approx 2.0$  Å. Neglecting the polar space group  $C_{\bullet,\bullet}^0$ , these conditions are fulfilled only if 4 (O, F) atoms, here called  $(O, F)_1$ , are situated at the positions  $\pm 0.02$  with  $z \approx 0.12$ , and 4 (O, F) atoms,  $(O, F)_2$ , at the positions  $0 \cdot 10$ ,  $10 \cdot 10$  Assuming the distance (O, F)-(O, F) to be  $\geq 2.5$  Å and the distance Bi-(O, F) to be  $\geq 2.2$  Å there is only room for the remaining 4 (O, F) atoms,  $(O, F)_3$ , at the positions  $0 \cdot 12$ ,  $10 \cdot 12$ .

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I <sub>calc</sub> 37 3 0.05 34 6	VW.		0 2 4 6 8		Icalo 210 17 11 110 27	Ioba vst w w m	I <sub>calc</sub> 36 4 2 33	Ios w vvw — w vw
2	- :		10 12		2 21	vvw `m	1 20	
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1								
13		8	vw	3	vw	į		
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either from the arity in the reefore treated as ae O and the F F for which z<sub>Bi</sub>

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end by a regular  $(O, F) \approx 2.0 \text{ Å}$ . only if 4 (O, F) th  $z \approx 0.12$ , and listance (O, F)—is only room for

The coordination and distances in A will be:

$$\begin{array}{lll} \text{Bi-4 (O, F)}_3 = 2.29 & \text{Nb-2 (O, F)}_1 = 2.0 \\ \text{Bi-4 (O, F)}_1 = 2.9 & \text{Nb-4 (O, F)}_2 = 1.92 \\ \text{(O, F)}_3 - 4 \text{(O, F)}_3 = 2.71 & \text{(O, F)}_2 - 4 \text{(O, F)}_2 = 2.71 \\ \text{(O, F)}_1 - 4 \text{(O, F)}_3 = 2.9 & \text{(O, F)}_1 - 4 \text{(O, F)}_2 = 2.8 \end{array}$$

The proposed structure is given in the summary. Calculated and observed intensities for the reflexions in the Weissenberg photographs are given in Table 3.  $I_{calc}$  is derived as follows:

$$I_{\text{calc}} = \frac{1 + \cos^2 2\theta}{1600 \cdot \sin 2\theta} \cdot F^2 \text{ where } F = \sum f \cos 2\pi (h x + k y + l z).$$

The lines of maximum absorption in the Weissenberg photographs (see 5) are indicated by dotted lines in Table 3. If the absorption effects are taken into account, the agreement between calculated and observed intensities is quite good.

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### Discussion of the structure

Even if the (O, F), atoms (see above) are assumed to occupy such a position that the distance  $Bi-4(O, F)_1$  is as small as possible  $[z(O, F)_1=0.15, (O, F)_1=0.15]$  $(O, F)_8 = 2.5, Nb-2 (O, F)_1 = 2.5, Bi-4 (O, F)_1 = 2.7$ , the distance  $Bi-4 (O, F)_1 = 2.7$ still be much longer than the distance Bi-4 (O, F)3. It therefore seems an propriate to describe the structure of Bi<sub>2</sub>NbO<sub>5</sub>F as being built up of Bi<sub>2</sub> (O, Fig. layers alternating with octahedral layers having the composition Nb (O, F)4, both layers being perpendicular to the c-axis. It might be pointed out that the structure of  $Bi_2(0, F)_2 Nb(0, F)_4$  is basically of the same type as the " $X_1$  structures",  $Me_2O_2X_1$ previously investigated by Sillen (6). Thus the octahedral sheets Nb(O, F), correspond to single layers of halogen atoms, X, in  $Me_2O_2X$ .

### Discussion of the distribution of the O and the F atoms

As seen above, the distance Bi-4(O, F)<sub>3</sub> is 2.29 A for Bi<sub>2</sub>NbO<sub>5</sub>F. For Bi<sub>2</sub>TiO<sub>4</sub>F<sub>5</sub> the corresponding distance is calculated to be 2.26 ± 0.06 Å. These distances are very nearly the same as the corresponding distances, Bi-4 O, within the Bi<sub>2</sub>O<sub>4</sub> layers of other bismuth oxicompounds (6). This need not, however, necessarily mean that the Bi<sub>2</sub>(O, F)<sub>2</sub> layers (see the figure) are free from F atoms, since compounds with  $Bi_2(O, F)_2$  layers which certainly contain F atoms have not been investigated hitherto, and thus the distances within such layers are unknown.

For the present it seems therefore best to make no special assumptions as to the distribution of the O and F atoms.

### SUMMARY

The crystal structure of Bi<sub>2</sub>NbO<sub>5</sub>F has been investigated by means of Weissenberg and powder photographs. From powder photographs the compounds Bi<sub>2</sub>TaO<sub>5</sub>F and Bi<sub>2</sub>TiO<sub>4</sub>F<sub>2</sub> have been found to be isomorphous with Bi<sub>2</sub>NbO<sub>5</sub>F. The following structure is proposed for Bi<sub>2</sub>NbO<sub>5</sub>F:

```
D_{4h}^{17}-I_{4}/mmm
            (000, \frac{1}{2}, \frac{1}{2}, \frac{1}{2}) +
   2 Nb in 2 (a):000
   4 Bi in 4 (e): \pm 00z
                                                z = 0.325 \pm 0.001
4(0, F)_1 in 4(c): 010, 100
4 (0, \mathbf{F})_{3} \text{ in } 4 (e) : \pm 00z
                                               z=0.12\pm0.01
4(0, F)_3 in 4(d):011, 101
```

The cell dimensions are a=3.835 Å, c=16.63 Å for  $Bi_2NbO_5F$ . The positions of the metal atoms were determined from the diffraction data, those of the (O, F) atoms from space considerations. Although it does not seem improbable that O atoms alone occupy the positions  $0\frac{1}{2}$ ,  $\frac{1}{2}0\frac{1}{2}$ , Bi and O thus forming  $Bi_2O_2$  layers as in other bismuth oxicompounds, nothing can be definitely stated as to the distribution of the O and F atoms.

The structure is built up of quadratic Bi<sub>2</sub>(O, F)<sub>2</sub> layers alternating with octahedral sheets having the composition Nb(O, F)4 (see figure) and the formula might thus be written: Bi<sub>2</sub>(O, F)<sub>2</sub>Nb(O, F)<sub>4</sub>. The structure is formally related to a series of ds

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previously investigated structures of general formula  $Me_2O_2$  ( $Me'_{m-1}R_mO_{3m+1}$ ) and represents the simplest case of this series, i.e. m=1.

University of Stockholm, Institute of Inorganic and Physical Chemistry, May 1952.

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### toms.

. For Bi<sub>2</sub>TiO<sub>4</sub>F<sub>3</sub> se distances are rithin the Bi<sub>2</sub>O<sub>2</sub> necessarily mean ince compounds een investigated

such a position

=0.15,  $(O, F)_1$ -4 3i-4  $(O, F)_1$  will

efore seems ap. ip of Bi<sub>2</sub> (O, F).

Nb (O, F)4, both

tures",  $Me_2O_2X$ , O, F), correspond

nat the structure :

ptions as to the

s of Weissenberg s Bi<sub>2</sub>TaO<sub>5</sub>F and following struc-

The positions of the (O, F) that O atoms Bi<sub>2</sub>O<sub>3</sub> layers as to the distribu-

with octahedral rula might thus I to a series of

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